Conference Paper

Microstructure Features of Ternary Alkali-activated Binder Based on Tungsten Mining Waste, Slag and Metakaolin

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Abstract
This study determines the effect of ground granulated blast furnace slag (GGBFS) and metakaolin (MK) on the microstructural properties of the tungsten mining waste-based alkali-activated binder (TMWM). During this investigation, TMWM was partially replaced with 10 wt.% GGBFS and 10 wt.% MK to improve the microstructure of the binder. In order to understand the effect of the substitutions on the microstructure, two pastes were produced to make a comparative study between the sample containing 100% TMWM and the ternary precursors. Both precursors were activated using a combination of alkaline activator solutions (sodium silicate and sodium hydroxide) with the ratio of 1:3 (66.6 wt.% sodium silicate combined with 33.33 wt.% of NaOH 8M). The alkali-activated mixes were cured in oven at temperature of 60 °C in the first day and at room temperature for the next 27 days. The reaction products N-A-S-H gel and (N,M)-A-S-H gel resulted from the alkaline activation reaction process. In addition, a formation of natrite (Na$_2$CO$_3$) with needles shape occurred as a reaction product of the fluorescence phenomena. However, a dense matrix resulted from the alkaline activation of the ternary precursors containing different gels such as N-A-S-H, C-A-S-H and (N,M)-C-A-S-H gel, these results were obtained through SEM-EDS analyses, as well FTIR tests.

Keywords: Mining Waste, Alkali-activated, Microstructure, Slag, Metakaolin

1. Introduction

The world nowadays facing a challenge to find solutions for the mitigation of waste accumulation on landfills and global warming increase. Recently, new clean technologies are applied as a potential solutions to mitigate global environmental threats by reusing the pollutant materials and sequestrate industrial waste and greenhouse gases [1]. The valorisation and re-using of some waste minerals to produce a new cementitious materials using the alkaline activation process technology by blend the precursors (minerals waste) with the alkaline activator solutions is one of the solution that allow to re-use such waste [2]. Those mineral waste are alumino-silicate rich materials (such
as mining waste [3], brick powder waste [4] and fly ash [5]) and others supplementary cementing materials such as by-products waste (Slags: like electric arc furnace slag [6] and GGBFS [7]). Moreover, some studies were using metakaolin (MK) which has been largely investigated for many years in the production of geopolymers due to its high reactivity and the homogeneous reaction products that formed after the geopolymerization [8]. Those materials are able to form a different reaction products such as: N-A-S-H, C-A-S-H and their hybrid (N,C)-A-S-H gels depend to the type of the precursors and the activator solutions used. The main alkaline activation product resulting from TMWM activation is N-A-S-H type gel. However, when TMWM blended with other waste such as glass waste powder formed more (C, N)-A-S-H gel. Where the Ca released by WG dissolution is incorporated into the N-A-S-H type gel [9].

The current study aimed to evaluate the feasibility of incorporating amount of GGBFS and MK into TMWM as a source of silica, alumina and calcium to enhance the microstructural properties of the tungsten mining waste based-alkali-activated binder and produce more alkaline activation products. In addition, the addition of GGBFS and MK was to improve the reactivity of the TMWM precursor. The microstructure of the 100%TMWM-AAB and the ternary-AAB (the blending of TMWM, GGBFS, MK and the activator solutions) were investigated using SEM-EDS and FTIR analyses.

2. Experimental Programme

2.1. Materials

2.1.1. Precursors

As received from Panasqueira mine located in Covilhã, Portugal. TMWM was dried in oven at 60 °C temperature for 24 hours. Afterwards the dried mud was mechanically disaggregated using a crushing machine and sieved under 500 µm particle size. The slag used in the current study is a ground granulated blast-furnace slag type derived from the production unit of the cement plant of Ain El Kebira (SCAEK), north-east of Algeria. The slag was grinded and sieved under 63 µm in order to use as a cement additive. While GGBFS is a by-product resulting from the separation of the metal from the ore at the metallurgic unit of El Hadjar located in the north-east of Algeria. The metakaolin (METAMAX) high reactivity metakaolin (HRM) with a strength activity index as per ASTM C-1240, supplied by the chemical company BASF Catalysts LLC.
The particle size distribution curves of milled TMWM, GGBFS and MK are shown in Fig. 1. The powders were determined using laser diffraction (CILAS 1190). The median particle size diameter \(D_{50}\) of TMWM, GGBFS and MK were 177.37 \(\mu m\), 14.83 \(\mu m\) and 3.8 \(\mu m\) respectively. Moreover, \(D_{10}\) of TMWM, GGBFS and MK were 8.04 \(\mu m\), 1.58 \(\mu m\) and 1.3 \(\mu m\) respectively. \(D_{10}\) is the diameter at which 10\% of the sample’s volume is comprised of particles with a diameter less than this value. In addition, \(D_{90}\) is the diameter at which 90\% of the sample’s volume is comprised of particles with a diameter less than this value. The \(D_{90}\) of TMWM= 398.6 \(\mu m\), GGBFS= 35.2 \(\mu m\) and for MK= 11.3 \(\mu m\).

The chemical compositions of the precursors are shown in Table 1 was determined by Hitachi S-4800 microscope instrument through the EDS (Energy Dispersive X-ray) analysis. The TMWM and MK contains high levels of SiO\(_2\) and Al\(_2\)O\(_3\), however, GGBFS contain high levels of SiO\(_2\) and CaO which are essential for alkali-activation, together with moderate amounts of Mg contain in TMWM and GGBFS.

### Table 1: Chemical composition of TMWM, GGBFS, MK and the ternary precursors.

<table>
<thead>
<tr>
<th>Component</th>
<th>TMWM</th>
<th>GGBFS</th>
<th>MK</th>
<th>Ternary Precursor</th>
</tr>
</thead>
<tbody>
<tr>
<td>Al(_2)O(_3)</td>
<td>17.05</td>
<td>7.79</td>
<td>43.2</td>
<td>18.74</td>
</tr>
<tr>
<td>SiO(_2)</td>
<td>46.67</td>
<td>34.1</td>
<td>51.4</td>
<td>45.89</td>
</tr>
<tr>
<td>TiO(_2)</td>
<td>0.6</td>
<td>0.37</td>
<td>2.5</td>
<td>0.77</td>
</tr>
<tr>
<td>SO(_3)</td>
<td>7.9</td>
<td>1.27</td>
<td>0.04</td>
<td>6.45</td>
</tr>
<tr>
<td>K(_2)O</td>
<td>4.9</td>
<td>0.51</td>
<td>0.16</td>
<td>3.99</td>
</tr>
<tr>
<td>CaO</td>
<td>0.69</td>
<td>48.7</td>
<td>0.3</td>
<td>5.45</td>
</tr>
<tr>
<td>Fe(_2)O(_3)</td>
<td>15.47</td>
<td>0.4</td>
<td>0.51</td>
<td>12.47</td>
</tr>
<tr>
<td>MgO</td>
<td>4.83</td>
<td>5.09</td>
<td>0.12</td>
<td>4.39</td>
</tr>
<tr>
<td>Na(_2)O</td>
<td>0.8</td>
<td>0.54</td>
<td>0.3</td>
<td>0.72</td>
</tr>
<tr>
<td>MnO</td>
<td>-</td>
<td>0.23</td>
<td>0.1</td>
<td>0.03</td>
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<tr>
<td>ZnO</td>
<td>1.09</td>
<td>-</td>
<td>0.1</td>
<td>0.88</td>
</tr>
</tbody>
</table>

### 2.1.2. Alkaline activators

Two types of alkaline activators were used in the current study, sodium silicate and sodium hydroxide (NaOH).

Sodium silicate (ref. D40) with the chemical composition: (8.60 wt.% Na\(_2\)O, 27.79 wt.% SiO\(_2\), 63.19 wt.% H\(_2\)O, and 0.4 wt.% Al\(_2\)O\(_3\)) with modulus of silicate \((\text{SiO}_2/\text{Na}_2\text{O}) = 3.23\), supplied by Solvay SA, Póvoa de Santa Iria, Portugal.
Sodium hydroxide flakes of 98.6% purity were supplied by DOS SANTOS, LDA Portugal. Sodium hydroxide solution was prepared by dissolving NaOH pellets in distilled water to provide concentration of 8M and allowed to cool 24 h before use.

![Figure 1: Particle size distribution curves of raw materials (TMWM, GGBFS and MK).](image)

### 2.2. Paste preparation and experimental techniques

Two alkali-activated binders were investigated by blinding the precursors with the alkaline activator solutions. The first precursor was 100% TMWM however the second precursor was a ternary material (80 wt.% TMWM + 10 wt.% GGBFS + 10 wt.% MK). The ternary precursors were mixed together in a container and agitated to ensure the homogeneity of the material. After that the precursors were activated with 66.6 wt.% sodium silicate solution + 33.3 wt.% 8M NaOH with “solid/liquid” ratio of 2.5. The sodium silicate solution was mixed together with NaOH using Magnetic stirrers “agimatic e selecta” for 5 min. The resulted alkaline activator solution was added to the precursors and mixed by hand between 5-10 min. Then, the pastes were cured in an oven at 60 °C for 24 h. After the experiment, the samples were removed from the oven and cooled to laboratory temperature. At 28 days of the reaction, the pastes were prepared for the different analyses. The initial ratios [(Si/Al), (Ca/Si), (K/Al), and (Na/Si)] of the mixtures are given in Table 2

For SEM-EDS analyses the samples were examined by a Hitachi S-4800 microscope instrument. The EDS (Energy Dispersive X-ray) analyses were using Bruker Xflash 5010 Cooled by Peltier resolution≤129eV of Mn Detector EDS instrument. The specimens were coated with gold.
Fourier Transform Infrared spectroscopy (FTIR) using the Nicolet iS10 FTIR Spectrometer (Thermo Scientific), Smart iTR accessory instrument in transmittance mode at frequencies (4000–600\text{cm}^{-1}) with diamond HATR crystal. The powders were samples that smashed using marble rod and sieved under 75 \mu m).

### 3. Results and Discussion

#### 3.1. SEM-EDS analyses

The microstructure analyses of the 100\% TMWM and the ternary (TMWM, GGBFS and MK) alkali-activated pastes are shown in Fig. 2 and Fig. 3 respectively. Although the sample ternary-AAB had 80 wt.% TMWM, the microstructure was different compared to the sample prepared using 100 TMWM. The matrix formation process in TMWM-AAB is different from that seen in the ternary-AAB which results in different reaction products.

In all samples (TMWM-AAB and ternary-AAB), gel structure was observed. In general, four major morphological features were observed, crystalline particles (such as quartz or muscovite or illite) with reaction product on the surface [Fig.2a (spot M1)], dense gel phases were it could not be asserted whether in these mixtures coexisted C-A-S-H and N-A-S-H phases or one hybrid (M,C)-A-S-H gel where M is Na or K, tabular type particles with no sharp geometric outline (unreacted particles) and sharp geometric needles [Fig. 2 (spot M2)]. In the TMWM-AAB the TMWM particles are well embedded and connected to the matrix, and some particles are partly covered with needle-like phase. The presence of the needles in the TMWM-AAB after the alkaline activation may correspond to the formation of natrite (Na$_2$CO$_3$) needles which are referred to [Fig. 2 spot M3]. These natrite (Na$_2$CO$_3$) needles does not appear in ternary-AAB SEM image, this may caused by the low reactivity of TMWM were the Na content in the mixture remained as unreacted and during the curing time the Na reacted with the CO$_2$ in the atmosphere and formed Na$_2$CO$_3$ under the efflorescence phenomena, as it shows in equation (1) and (2).

\[
\text{CO}_2(\text{g}) + 2\text{OH}^-_{(aq)} \rightarrow \text{CO}_2^{3-}_{(aq)} + \text{H}_2\text{O} \quad (1)
\]

\[
2\text{Na}^+_{(aq)} + \text{CO}_2^{3-} \text{(aq)} + 7\text{H}_2\text{O} \rightarrow \text{Na}_2\text{CO}_3.7\text{H}_2\text{O}_{(s)} \quad (2)
\]
According to Zhang et al. [10], this is therefore a partial neutralization for alkaline activation under natural carbonation conditions, as dissolved CO$_2$ acts as an acid and consumes hydroxides. The main reason for efflorescence in these materials is the availability of mobile Na$^+$ and OH$^-$, which are remains as unreacted from the alkaline activator solutions. However, in the Ternary-AAB the Na contend in the alkaline activator solution reacted with the precursors (TMWM, GGBF and MK) during the alkaline activation process, then there was no formation of natrite (needles structure) and it does not appear in ternary-AAB SEM image fig. 3b. With addition of 10 wt. % GGBFS and 10 wt. % MK to the TMWM, the change in microstructure was distinctly evident. In general, the microstructure of the ternary-AAB sample was denser due to increasing formation of almino-silicate reaction products and C-S-H gel. In fig 3.b the unreacted particles were found coated with the reaction product.

Table 3 shows the average content of the main elements (Si, Al, Na, Ca and K) token by SEM-EDS analysis, and their ratios in the reaction products in both alkali-activated mixtures. The values given in this table represent three different EDS spots for each alkali-activated sample as it shown in the SEM images. The differences in the precursors content and the main elements ratios as a function of the blend composition indicated the formation of a new reaction products with different compositions and structures. The ratios in the Table 3 were compared to the ratios set by Puligilla and Mondal [11] to conclude the type of the alkaline activation products that formed were: (M, N)-A-S-H: 0 < K/Al < 1.85; 0 < Ca/Si < 0.3; 0.05 < Al/Si < 0.43. K-A-S-H: K/Al = 0.47 ± 0.44; Ca/Si = 0.05 ± 0.03; Al/ Si = 0.23 ± 0.03.

Mainly two types of gel were identified in the tungsten mining waste mud based-alkali-activated binder, a gel with a high silicon and aluminium content and smaller amounts of sodium, potassium and calcium which the authors believe may be a N-A-S-H gel containing a certain amount of calcium (Ca) or potassium (K) and forming ((M,N)-A-S-H [spot M1]). However, based only on EDS analysis results, it could not be asserted whether in these mixtures coexisted C-A-S-H and N-A-S-H phases or one hybrid (M, N)-A-S-H gel as reported previously. In the presence of calcium, N-A-S-H gel evolved as follows: N-A-S-H → (N, C)-A-S-H → C-A-S-H. This last conversion is not complete in these systems because the amount of calcium present is thought to be insufficient [12]. Additionally, [spot M2] shows a high amount of silicon (18.54), aluminium (12.51) and sodium (13.36), with the ratios Al/Si= 0.675 and Na/Si= 1.068 and poor ratios of Ca/Si= 0.021 and K/Al= 0.048, which refers to the formation of N-A-S-H gel in the alkaline activation matrix. The EDS analysis of the needles observed in the SEM image [spot M3], confirmed the high concentration of sodium Na$^+$ (53.37 wt.%) and poor presence
of reactive elements such as calcium, aluminium and potassium had produced sodium carbonate (Natrite) by atmospheric carbonation. The analysis of the blend ternary-alkali-activated binder revealed that it contained an amounts of unreacted particles from the precursors (TMWM, Slag and MK) such as quartz with a high silicon (44.7 wt.%) content and trace of other composite and poor ratios of Al/Si= 0.015 and Ca/Si= 0.011 [spot T1], along with two alkaline activation products with different compositions: a gel with a high silicon and aluminium content and smaller amounts of sodium, potassium and calcium which the authors believe may be a N-A-S-H gel containing a certain amount of calcium and potassium ((M,N)-A-S-H [spot T2]). Another type of reaction product [spot T3] was formed with high content of silicon (25.76), calcium (9.69), aluminium (6.99) and sodium (4.72), and smaller amounts of potassium (0.84) with a high ratios Na/Al (0.675) and Ca/Si (0.376) were calcium readily displaces Na by ion exchange potential is envisaged for the surface of the 3D N-A-S-H gels leading to (N,C)-A-S-H gels. It is anticipated that high content of Ca in GGBFS will be adsorbed preferentially, and Na will only be retained if insufficient Ca is available [13].

Figure 2: Scanning electron microscope images of the TMWM-based alkali-activated binder.

Figure 3: Scanning electron microscope images of the Ternary alkali-activated binder.
TABLE 3: Element composition and ratios of the TMWM-AAB and Ternary-AAB pastes after the alkaline activation in different spots.

<table>
<thead>
<tr>
<th>Materials</th>
<th>Spots</th>
<th>Element (wt.%)</th>
<th>Ratios after activation</th>
<th>Phases</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td></td>
<td>Si</td>
<td>Al</td>
<td>Na</td>
</tr>
<tr>
<td>TMWM-AAB</td>
<td>Spot M1</td>
<td>23.27</td>
<td>10.67</td>
<td>6.6</td>
</tr>
<tr>
<td></td>
<td>Spot M2</td>
<td>18.54</td>
<td>12.51</td>
<td>13.36</td>
</tr>
<tr>
<td></td>
<td>Spot M3</td>
<td>3.27</td>
<td>3.02</td>
<td>53.37</td>
</tr>
<tr>
<td>Ternary-AAB</td>
<td>Spot T1</td>
<td>22.14</td>
<td>7.86</td>
<td>5.89</td>
</tr>
<tr>
<td></td>
<td>Spot T2</td>
<td>25.76</td>
<td>6.99</td>
<td>4.72</td>
</tr>
<tr>
<td></td>
<td>Spot T3</td>
<td>44.7</td>
<td>0.69</td>
<td>0.68</td>
</tr>
</tbody>
</table>

3.2. FT-IR analyses

Fig. 4 shows the FTIR spectra of the two alkali-activated binders, the first binder based on tungsten mining waste as precursor, however the second binder based on ternary (TMWM, GGBFS, MK) precursors. In FT-IR spectra, there was a peak at approximately 3647 cm⁻¹, which was found to be related to O-H stretching vibrations. Besides, this peak at the raw TMWM due to the presence of muscovite. In addition, there is broadband in the region of 3215-3353 cm⁻¹ and a peak 1648 cm⁻¹ refer to the spectrum of bending H-O-H vibrations and stretching O-H groups respectively[14]. These wavenumbers correspond the weakly bound water molecules which were adsorbed on the surface or trapped in the large cavities between the rings of the alkaline activation products. This bands were disappeared in the raw precursors (TMWM, GGBFS and MK) due to the absence of activation products in the raw materials before the alkaline activation process, and it belongs to the reaction products. Whereas a sharp peak centred approximately 2980 cm⁻¹ associated to the O-H stretching vibrations of bound water [15]. The stretching vibration of O-C-O bond at approximately 1450-1410 cm⁻¹ indicate the presence of carbonate mineral which having occurred by the atmospheric carbonation, and the coexist of calcite (CaCO₃) as unreacted particles from GGBFS. Moreover, the presence of natrite (Na₂CO₃) may due to the carbonation of the large amounts of Na carried to the surface. The peaks of TMWM-AAB and ternary-AAB show the same peaks intensity even though ternary-AAB sample contains 10 wt.% of GGBFS that has calcite. Then the TMWM-AAB formed more sodium carbonate due to the unreacted sodium content in the mixture then the unreacted sodium formed sodium carbonate with CO₂ of the atmosphere. which is seen before in the SEM
image corresponding to TMWM-AAB (Fig 2). In addition, a broad band at wavenumber 1436 cm\(^{-1}\) appeared only in the TMWM-AAB which correspond to sodium carbonate Na\(_2\)CO\(_3\) (Natrite) \([16]\), and disappeared in the ternary-AAB and the starting materials means that natrite (Na\(_2\)CO\(_3\)) formed only in TMWM-AAB after the alkaline activation. However, the ternary-AAB that does not show this band means that no natrite are formed and all the soluble sodium contains in the activator are reacted with the precursors. This explain why the natrite needles appears in the SEM images of TMWM-AAB and disappeared in the SEM images of T-AAB (SEM-EDS section). The intensity of the bands at wavenumbers 1245-1249 cm\(^{-1}\) and at wavenumber 1073 cm\(^{-1}\) attributed to the length as well as the angle of the Si-O and Al-O bonds in the silicates and MK, in this case being sodium alumino-silicate (N-A-S-H) and/or calcium alumino-silicate (C-A-S-H) and probably other XRD-amorphous silicate phases \([17]\). A wide and intense band appears from wavenumber 950 cm\(^{-1}\) to 1200 cm\(^{-1}\). The interpretation of this region may be difficult, as bond vibrations of different compounds tend to overlap from the raw precursors and also the samples after the alkaline activation, producing a highly complex spectrum. While in the unreacted TMWM this band appears at wavenumber 982 cm\(^{-1}\). The FTIR spectra for the pastes attributed to the asymmetric stretching vibrations generated by T-O-T bonds (where T is Si or Al) \([18]\). The band generated by the SiO\(^4\) groups in the TMWM slightly shifted from 982 cm\(^{-1}\) to 980 cm\(^{-1}\) for the sample made only by TMWM as precursor. However, the band considerably shifted from 982 cm\(^{-1}\) to 966 for the sample made by the ternary precursors, due to the formation of alumino-silicate gels such as N-A-S-H and C-A-S-H-types gel. More C-A-S-H type of gel formed in the ternary alkali-activated binder, as a result of the alkaline activation process of GGBFS and MK, the Al-O asymmetric stretch band shifts towards lower frequencies at around 966 cm\(^{-1}\) (Ternary-AAB sample), indicating the formation of an alkaline alumino-silicate gel, due to the presence of calcium and reactive alumina and silica in GGBFS and MK respectively. The peak centred at 874 cm\(^{-1}\) correspond to the C-O bending vibration confirms the presence of CaO in the raw GGBFS. The peak at this wavenumber was disappeared after the alkaline activation of the ternary precursors (contains GGBFS in its composition), due to the reaction of CaO in the system.

The bands at 797 cm\(^{-1}\) and 770 cm\(^{-1}\) are respectively associated to the Si-O and Si-O-Si molecular structure vibrations correspond to the presence of quartz in the original TMWM. Although these bands persist after the activation of the precursors, a reduction in the peaks intensities is observed in the sample contains 100% TMWM, which indicates that the quartz has partially reacted. However, the peaks at the same wavenumber disappeared in the ternary alkali-activated binder spectra which indicates that more
amount of quartz has been partially reacted compared to TMWM-AAB sample. Moreover, the absorption peak at 694 cm\(^{-1}\) is due to Si-O-Si symmetrical stretching vibrations, which can be associated to the presence of quartz in the raw material TMWM.

![FTIR spectra](image)

**Figure 4**: FTIR spectra for the raw TMWM, GGBFS and the samples TMWM-AAB and Ternary-AAB. (a) In the 4000–500 cm\(^{-1}\) range. (b) In the 1500–500 cm\(^{-1}\) range.

### 4. Conclusions

Tungsten mining waste-based alkali-activated binder and ternary (TMWM+GGBFS+MK) alkali-activated binders have been synthesised in order to perform a detailed comparison about their composition and microstructure. These different alkali-activated binders have been studied to highlight differences and similarities in their microstructure.
Under the conditions of the present study, it appears that addition of GGBFS and MK to the TMWM improved the microstructure properties by forming a homogeneous matrix structure. More specific, the main reaction products obtained in tungsten mining waste mud based-alkali-activated binder were N-A-S-H gel, (M,N)-A-S-H gel and C-S-H, and also the formation of natrite (Na$_2$CO$_3$) needles caused by florescence phenomena. However, complex types of reaction products formed in the ternary-alkali-activated binder such as N-A-S-H gel, C-A-S-H gel, (N,C)-A-S-H gel, (M,N)-A-S-H gel and C-S-H gel. In addition, SEM-EDS and FT-IR analyses confirm the formation of a new product as natrite (Na$_2$CO$_3$) in the TMWM-AAB were it appeared as needles in the SEM image and a spot with high Na content in EDS spot. Additionally, FT-IR spectra shows the stretching vibration of O-C-O which confirm the finding of SEM-EDS analyses about the formation of the natrite in the TMWM-AAB.

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References


